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Optical spectroscopy of excited exciton states in MoS<sub>2</sub> monolayers in van der Waals heterostructuresC. Robert,<sup>1</sup> M. A. Semina,<sup>2</sup> F. Cadiz,<sup>1,3</sup> M. Manca,<sup>1</sup> E. Courtade,<sup>1</sup> T. Taniguchi,<sup>4</sup> K. Watanabe,<sup>4</sup> H. Cai,<sup>5</sup> S. Tongay,<sup>5</sup> B. Lassagne,<sup>1</sup> P. Renucci,<sup>1</sup> T. Amand,<sup>1</sup> X. Marie,<sup>1</sup> M. M. Glazov,<sup>2</sup> and B. Urbaszek<sup>1</sup><sup>1</sup>Université de Toulouse, INSA-CNRS-UPS, LPCNO, 135 Av. Rangueil, 31077 Toulouse, France<sup>2</sup>Ioffe Institute, 194021 St. Petersburg, Russia<sup>3</sup>Physique de la matière condensée, Ecole Polytechnique, CNRS, Université Paris Saclay, 91128 Palaiseau, France<sup>4</sup>National Institute for Materials Science, Tsukuba, Ibaraki 305-0044, Japan<sup>5</sup>School for Engineering of Matter, Transport and Energy, Arizona State University, Tempe, Arizona 85287, USA

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The optical properties of MoS<sub>2</sub> monolayers are dominated by excitons, but for spectrally broad optical transitions in monolayers exfoliated directly onto SiO<sub>2</sub> substrates detailed information on excited exciton states is inaccessible. Encapsulation in hexagonal boron nitride (hBN) allows approaching the homogenous exciton linewidth, but interferences in the van der Waals heterostructures make direct comparison between transitions in optical spectra with different oscillator strength more challenging. Here we reveal in reflectivity and in photoluminescence excitation spectroscopy the presence of excited states of the *A* exciton in MoS<sub>2</sub> monolayers encapsulated in hBN layers of calibrated thickness, allowing us to extrapolate an exciton binding energy of  $\approx 220$  meV. We theoretically reproduce the energy separations and oscillator strengths measured in reflectivity by combining the exciton resonances calculated for a screened two-dimensional Coulomb potential with transfer matrix calculations of the reflectivity for the van der Waals structure. Our analysis shows a very different evolution of the exciton oscillator strength with principal quantum number for the screened Coulomb potential as compared to the ideal two-dimensional hydrogen model.

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Two-dimensional (2D) crystals of transition metal dichalcogenides such as MX<sub>2</sub> (M=Mo, W; X=S, Se, Te) are promising atomically thin semiconductors for applications in electronics and optoelectronics [1–6]. The optical properties of transition metal dichalcogenides (TMD) monolayers (MLs) are governed by very robust excitons [7] with a binding energy of the order of 500 meV [8–14]. The interplay between inversion symmetry breaking and strong spin-orbit coupling in these MLs results in unique spin/valley properties [15–22].

The first reported optical spectroscopy measurements on ML TMDs were performed on the material MoS<sub>2</sub> [3,4], motivated by the high natural abundance of the naturally occurring mineral molybdenite [23]. But due to the spectrally narrower emission lines of ML MoSe<sub>2</sub> and WSe<sub>2</sub> [20,24,25], low temperature optical spectroscopy studies rapidly switched to other synthesized materials of the TMD family [24,26–33]. Well-defined optical transitions have allowed the observation of excited exciton states, and hence the extrapolation of exciton binding energies, in the optical spectrum of WSe<sub>2</sub> [8,14] and of WS<sub>2</sub> MLs [10,34–36]. Knowledge of the energy of the exciton resonances is also crucial for linear and nonlinear optics as in second-harmonic generation and also resonant Raman scattering [14,37–42].

Optical spectroscopy experiments on excited exciton states plays a crucial role aiming to distinguish between the *A*-exciton and the *B*-exciton Rydberg series and other excitonic transitions possibly involving carriers from different valleys in momentum space away from the *K* point. For example the  $\Gamma$  point in the valence band of MoS<sub>2</sub> is situated between the *A*- and *B*-valence spin-orbit bands in MoS<sub>2</sub> [43–45] according

to atomistic calculations and angle resolved photo-electron spectroscopy.

Due to the very broad optical transition with a linewidth for MoS<sub>2</sub> of up to 50 meV at low temperature when studied without hexagonal BN (hBN) encapsulation [18,19,46–50], information on excited exciton states in ML MoS<sub>2</sub> is scarce. Hill *et al.* [36] report the presence of excited states in the photoluminescence excitation spectroscopy (PLE) spectrum of ML MoS<sub>2</sub> at room temperature. The authors ascribe the observed resonances to the first excited states of the *B* exciton, and estimate an exciton binding energy of about 400 meV for monolayers deposited onto fused silica substrates. The excited states of the *A* exciton are predicted to be close in energy to the *B* exciton and therefore are not visible in samples with broad transitions studied so far.

Here we present the first measurements of the *excited* exciton states in encapsulated monolayer MoS<sub>2</sub> in reflectivity and PLE spectra, using the same encapsulation technique that resulted in ground state exciton transitions of down to 2 meV linewidth [51]. In our samples we can spectrally separate optical transitions stemming from the excited *A*-exciton states from the *B*-exciton *1s* state. We show in reflectivity measurements and simulations that the thickness of each layer of the van der Waals structure impacts the visibility of the exciton states. We discuss the deviations of the relative oscillator strengths and exciton binding energies of the observed Rydberg series for exciton transitions from the ideal 2D hydrogen problem and extrapolate an exciton binding energy of  $\approx 220$  meV. Moreover, we examine the possible role of optical transitions away from the *K* point of the Brillouin zone. We show efficient valley

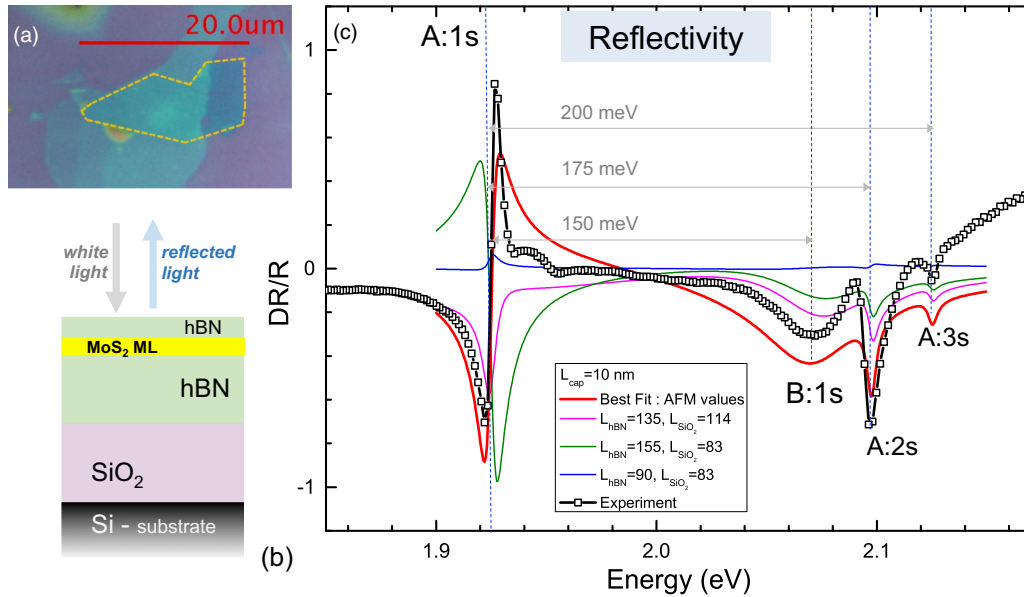


FIG. 1. Optical spectroscopy results. Sample 1. (a) Optical microscope image of an MoS<sub>2</sub> flake encapsulated in hBN. The scale bar is 20  $\mu\text{m}$ . (b) Schematic of sample structure, the top (bottom) hBN layers are  $7 \pm 3$  nm ( $130 \pm 5$  nm) thick as determined by AFM, the SiO<sub>2</sub> layer is 83 nm thick. (c) Differential reflectivity measurement ( $R_{ML} - R_{sub}$ )/ $R_{sub}$  performed with a power-stabilized white halogen lamp. The strong transition at lower energy is due to the *A*-exciton ground state absorption. At higher energies, three more transitions are clearly visible; the broader one is due to the *B*-exciton ground state while the other two are tentatively ascribed to be the first two excited states of the *A* exciton: *A* : 2*s* and *A* : 3*s*. Model simulations using the hBN thicknesses as determined by AFM are shown by the bold, red curve. Additional simulations (fine magenta, green, and blue) for different hBN and SiO<sub>2</sub> thicknesses show how the depth and shape of the exciton resonances depends on the individual layer thicknesses. The exciton resonance parameters are as follows: The energy positions of *A* : 1*s* exciton were tuned to the observed *A* : 1*s* peak; energies of excited states are calculated and shown in Fig. 3(a); radiative damping for the ground states  $\Gamma_{0,A:1s} = \Gamma_{0,B:1s} = 1$  meV, for excited states found from calculation, Fig. 3(b); nonradiative damping  $\Gamma_A = 2.5$  meV,  $\Gamma_B = 25$  meV. Refractive indices:  $n_{\text{hBN}} = 2.2$ ,  $n_{\text{SiO}_2} = 1.46$ ,  $n_{\text{Si}} = 3.5$ .

polarization and coherence initialization for laser energies tuned into resonance with the excited *A*-exciton states.

**Experimental Results.** We have investigated MoS<sub>2</sub> ML encapsulated in hBN on top of SiO<sub>2</sub>/Si substrate, see optical image in Fig. 1(a). In atomic force microscopy (AFM) measurements we determined that the top hBN layer thickness is  $7 \pm 3$  nm, the bottom hBN  $130 \pm 5$  nm thick. The SiO<sub>2</sub> layer is 83 nm thick. These van der Waals heterostructures are obtained by mechanical exfoliation of bulk MoS<sub>2</sub> (from 2D Semiconductors, USA and grown by chemical vapor transport as in Ref. [51]) and hBN crystals [52], following the fabrication technique detailed in Ref. [51]. Encapsulation results in high optical quality samples with well defined optical transitions (FWHM < 5 meV) both in photoluminescence (PL) and reflectivity at low temperature, as recently shown [51,53–56]. The PL linewidth of the neutral exciton ( $X^0$ ) thus reaches values down to 2 meV at cryogenic temperatures, comparable to high quality III-V and II-VI quantum wells grown by molecular beam epitaxy emitting at similar wavelength [57–59]. These well-defined emission lines are critical for an in-depth analysis of the optical transitions, since the narrow exciton lines allow us to clearly identify transitions involving the *A*-exciton excited states as they can be spectrally separated from the *B* : 1*s* exciton. Here the states are denoted by, e.g., *A* : *n**s* with *n* being the principal quantum number, in analogy to the *s*-shell (zero angular momentum) states of the hydrogen atom, small mixing of the *s*- and *p*-shell excitons expected in MX<sub>2</sub> MLs [42,60] is neglected here for simplicity.

Figure 1(c) presents the reflectivity spectrum at  $T = 5$  K of an encapsulated ML, in which we clearly observe the peaks corresponding to the absorption of the lowest energy transition of the *A* : 1*s* and the *B* : 1*s* excitons at 1.926 eV and 2.08 eV, respectively. The  $\sim 150$  meV energy separation between the *A* and *B* exciton is in agreement with previous measurements [17,61] and reflects mostly the spin-orbit splitting of the valence band at the *K* points of the hexagonal Brillouin zone [44,62,63]. In addition, higher energy states with measurable oscillator strengths are also visible, that we tentatively ascribe to the two first excited states of the *A* exciton: *A* : 2*s* and *A* : 3*s*.

These new features above the *B* : 1*s* transition shown in Fig. 1(c) deserve a detailed analysis. We present now the key results associated to the resonant excitation of the excited exciton states in PL excitation (PLE) with a tunable laser source, see Supplemental Material [64] for experimental details. We tune the laser into resonance with the *A* : 2*s* transition at 2.1 eV. The resulting PL of the *A* : 1*s* state is strongly co-polarized with the laser as shown in Fig. 2(a). For linearly polarized excitation  $\sigma^X$ , the resulting *A* : 1*s* PL is linearly polarized as a consequence of the optical generation of valley coherence, i.e., optical alignment of excitons [20,25,33]. For circularly polarized laser excitation  $\sigma^+$ , the PL is co-circularly polarized indicating efficient valley initialization. The fact that a relatively strong polarization is obtained for an excitation laser energy of  $\approx 175$  meV above the neutral exciton transition is a consequence of a resonance with the *A* : 2*s* excited state, which

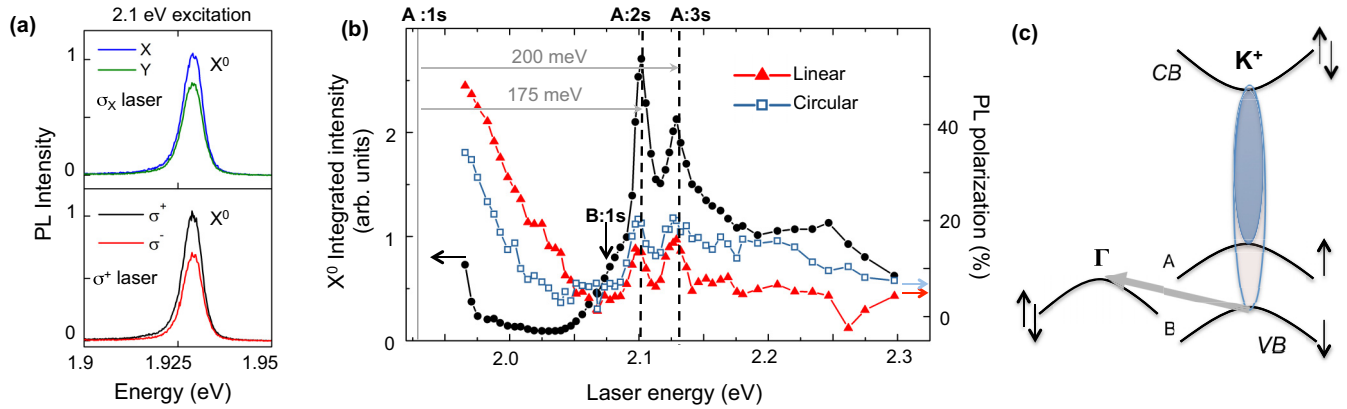


FIG. 2. Sample 2. (a) Polarization-resolved photoluminescence at  $T = 4$  K following linear (top) and circular (bottom) excitation at 2.1 eV, exhibiting efficient valley coherence and valley initialization, respectively. (b) PLE measurements. The integrated  $X^0$  (i.e., transition  $A : 1s$ ) intensity as a function of excitation energy is shown (black circles), as well as the linear (resp. circular) polarization obtained under linear (resp. circular) excitation (red triangles, open squares). (c) Schematic single particle band structure of ML  $\text{MoS}_2$ , for simplicity the small spin splitting in the conduction band (CB) is neglected.

relaxes efficiently down to the  $A : 1s$  state [65]. As shown in Fig. 2(b), the  $A : 1s$  integrated intensity as a function of excitation energy exhibit two clear and sharp resonances at the  $A : 2s$  and  $A : 3s$  energies. Remarkably, the linear (circular) polarization of the  $A : 1s$  emission under linearly (circularly) polarized excitation also exhibit local maxima when in resonance with the excited  $A$ -exciton states. This points towards a fundamental connection between the  $A : 1s$  and these peaks and is an argument in favor of attributing the two transitions to the  $A : 2s$  and  $A : 3s$  states. Both valley coherence and valley polarization reach very high values of  $\sim 40\%$  when approaching with the laser the resonance energy of the  $A : 1s$  transition. Interestingly, the degree of linear polarization is even higher than the circular polarization for excitation energies below 2.05 eV, as recently observed for nearly resonant excitation [51]. This is expected if the exciton spin/valley relaxation process is dominated by Coulomb exchange interaction [66].

Although the  $B : 1s$  state is clearly visible in reflectivity, it is much less pronounced in PLE, where  $B : 1s$  appears as a low energy shoulder of the  $A : 2s$  transition. A strong signal in PLE relies on efficient absorption at the excitation energy but also on efficient relaxation from this energy to the  $A : 1s$  state, involving phonon emission [56,67,68]. The weak response in PLE of the  $A : 1s$  state at the  $B : 1s$  energy might be due to inefficient relaxation to the  $A : 1s$  state, with the  $\Gamma$  valence states providing an alternative relaxation channel for holes, as sketched in Fig. 2(c). This is probably because the energy splitting between the valence  $K$  and  $\Gamma$  states is only about 100 meV [43–45]. Fast  $B$ -exciton relaxation could also contribute to the spectral broadening of the transition.

To confirm that the transitions we uncover in PLE and reflectivity can be ascribed to neutral and not charged excitons, we have performed experiments on gated structures [69,70], see Supplemental Material [64], where we also present results on up-conversion and hot PL.

**Discussion.** The fascinating optical properties of TMDC MLs are based mainly on the transitions at the  $K$  point of the Brillouin zone. Here we try to explore optical transitions at higher energy than the exciton ground state. Optical transitions at higher energy can be divided into three categories which

might spectrally overlap: (1) excited states of the  $A$  exciton, (2) ground state  $B$  exciton,  $B : 1s$ , (3) other transitions in the Brillouin zone, possibly indirect and therefore phonon assisted. The presence of transitions in reflectivity demonstrates nonzero oscillator strength for the absorption, which indicates that phonon-assisted processes are most likely not at the origin of the transitions. Moreover, the PLE data presented above demonstrates a close relation between the higher energy peaks and the ground  $A : 1s$  exciton states and gives arguments in favor of ascribing the sharp peaks in the reflectivity to the excited states of  $A$  exciton. Below we provide quantitative analysis of the energies and oscillator strengths of the  $A$ -exciton Rydberg series and demonstrate its consistency with experimental observations.

We performed numerical calculations in order to estimate the exciton binding energy in these encapsulated monolayers. The Coulomb interaction in an encapsulated 2D material is modelled using the potential [71–75]

$$V(r) = -\frac{e^2}{8\epsilon_0 r_0} \left[ \mathbf{H}_0\left(\frac{\kappa r}{r_0}\right) - Y_0\left(\frac{\kappa r}{r_0}\right) \right]. \quad (1)$$

Here  $e$  is the electron charge,  $\epsilon_0$  is the vacuum permittivity,  $\mathbf{H}_0$  and  $Y_0$  are the Struve and Neumann functions, respectively, and  $r_0$  is a screening length characterizing the  $\text{MoS}_2$  dielectric nature. We then solved the Schrödinger equation with the potential in Eq. (1) by modeling the encapsulation by hBN by an effective relative dielectric constant  $\kappa = 4.5$  as in Ref. [76], and by using an exciton reduced mass of  $\mu = 0.25 m_0$  [74]. We disregard frequency dependence of  $\kappa$  and  $r_0$  because the exciton binding energy exceeds the phonon energies and correspondingly we use the high-frequency value of  $\kappa$  for hBN. The calculation results are shown in Fig. 3(a). For a screening length of  $r_0 = 2.95 \pm 0.1$  nm, we obtain the exciton binding energy of  $\approx 222$  meV, with the  $A : 1s - A : 2s$  separation of  $\approx 174$  meV and the  $A : 3s - A : 2s$  separation of  $\approx 28$  meV, in excellent agreement with the values observed in our reflectivity and PLE spectra ( $173 \pm 5$  and  $28 \pm 3$  meV, respectively). The screening length is related to the 2D polarizability via

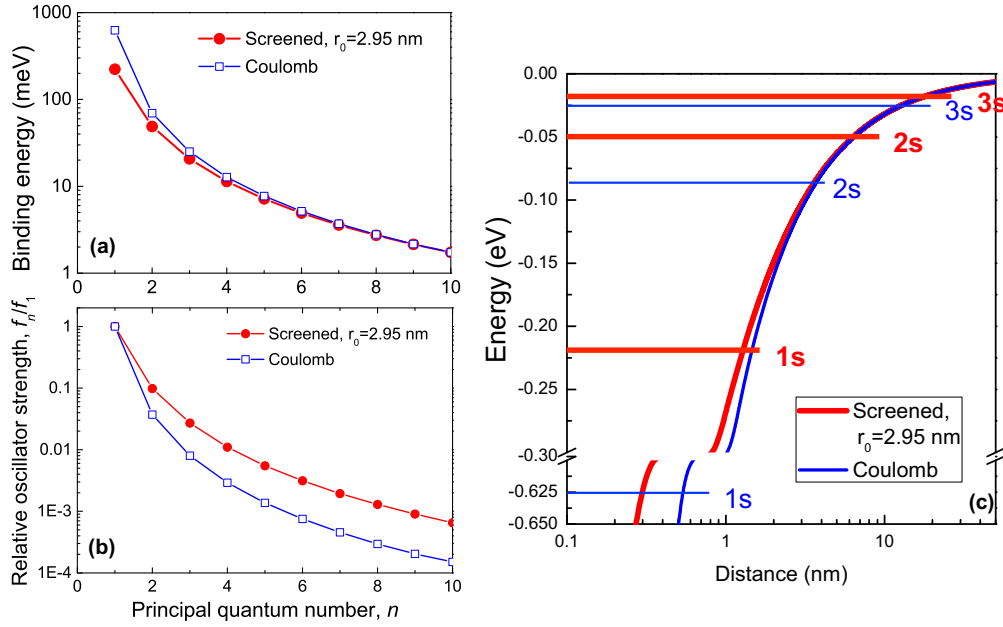


FIG. 3. Results of calculations. (a) Exciton binding energy for screened  $r_0 = 2.95$  nm, Eq. (1) and unscreened 2D Coulomb potential as a function of principal quantum number  $n$  for ML MoS<sub>2</sub>. (b) Same as (a), but for the relative oscillator strength normalized at the  $A : 1s$  exciton oscillator strength. (c) Comparison of the bound states in the screened and unscreened Coulomb potential, see Supplemental Material for more details [64].

$r_0 = 2\pi\chi_{2D}$ , corresponding therefore to  $\chi_{2D} = 4.47 \pm 0.15$  Å, in reasonable agreement with theory [74].

Deviations of the electron-hole interaction potential from the  $1/r$  law give rise also to the deviations of the oscillator strengths of  $A : ns$  states,  $f_n$ , from the ideal 2D exciton model, where  $f_n = f_1/(2n - 1)^3$  [77]. This is illustrated in Fig. 3 where the ratio  $f_n/f_1$  is plotted showing strong increase of the relative oscillator strengths for the screened interaction as compared to the standard Coulomb potential. In fact, the ground and first excited states are formed, as shown in Fig. 3(c), in a much shallower,  $\propto \log(r/r_0)$  effective potential, see Supplemental Material [64] for details. Hence, the oscillator strength, proportional to the probability to find the electron and the hole within the same unit cell, is smaller for the screened than for the ideal Coulomb potential. The higher the principal quantum number  $n$  is, the closer is  $V(r)$ , Eq. (1), to its  $1/r$  asymptotics. As a result the oscillator strengths approach those in  $1/r$  potential. Therefore, the decrease of  $f_n/f_1$  is weaker than for the ideal 2D exciton due to smaller oscillator strengths for the low  $n$  states. Note that comparatively strong excited exciton state resonances are also predicted from DFT-BSE calculations of the optical response of ML MoS<sub>2</sub>, as for instance in Ref. [12]. However, in these calculations the exact physical origin of a resonance in absorption is difficult to trace back to a precise quantum number.

Although the calculations demonstrate that the relative oscillator strengths for  $2s$  and  $3s$  excitonic states are higher than for the  $1/r$  unscreened potential, the amplitude of the  $A : 2s$  and  $A : 3s$  resonances in the reflectivity spectra are unexpectedly high and, at first glance, cannot be accounted for by the increased  $f_n/f_1$  in Fig. 3(b). However, the light reflection from our van der Waals sample is determined not only by the monolayer itself, but also by the hBN, SiO<sub>2</sub>, and Si layers, cf. Refs. [78,79]. The cap and bottom layers form a

microcavity-like structure enhancing the reflection of the light. To account for the details of light propagation in the sample we employed standard transfer matrix technique, see Refs. [64,80] for details and calculated the reflection contrast spectra for the studied sample. The amplitude reflection coefficient of MoS<sub>2</sub> ML was taken in the form

$$r(\hbar\omega) = \sum_{n=1}^3 \frac{i\Gamma_{0,A:ns}}{E_{A:ns} - \hbar\omega - i(\Gamma_{0,A:ns} + \Gamma_A)} + \frac{i\Gamma_{0,B:1s}}{E_{B:1s} - \hbar\omega - i(\Gamma_{0,B:1s} + \Gamma_B)}. \quad (2)$$

It includes independent contributions of the  $A : 1s$ ,  $2s$ , and  $3s$  excitons as well as that of the  $B : 1s$  exciton,  $E_{A:ns}$ ,  $E_{B:1s}$  are the corresponding energies,  $\Gamma_0$  and  $\Gamma$  are the radiative and nonradiative dampings of the excitons. Equation (2) assumes that the exciton resonances are well separated. The interference-enhanced reflection indeed improves the visibility of the excited states and the results of simulation plotted by the red curve in Fig. 1 reproduce all features of experimental data rather well. In the calculations the only fitting parameters where absolute energy positions of  $A : 1s$  and  $B : 1s$  excitons, its nonradiative dampings and radiative damping of the  $A : 1s$  exciton. The radiative damping of  $B : 1s$  exciton was set to be equal to  $\Gamma_{0,A:1s}$ . All other energies and radiative dampings were found from the calculations presented in Fig. 3. Note that small deviations of the layer thicknesses from the values found in AFM studies (green, blue, and magenta curves) result in completely different amplitudes and shapes of features in the reflectivity. This opens the way to control and engineer the optical spectra of the van der Waals heterostructures by choosing appropriate thicknesses of hBN and SiO<sub>2</sub> layers resulting in enhancement or suppression of excitonic resonances. We stress that while higher relative oscillator strengths for the excited

states as compared with the two-dimensional hydrogenic model is a general property of atom-thin monolayers due to the screened potential (1), the intensity of the features related to the excited states in the measured reflection spectra can strongly depend on the surrounding layers.

As additional identification, magneto-optics in high magnetic fields is desirable to check if the transitions assigned to different exciton states have a different diamagnetic shift, as recently demonstrated in magnetotransmission experiments on  $A : 2s$  to  $4s$  states in ML WSe<sub>2</sub> [76]. For a quantitative analysis of the oscillator strength of the excited exciton states in the experiment, in addition the impact of mixing of  $s$ - and  $p$ -shell excitonic states needs to be investigated [42,60].

In conclusion direct measurements of the excited states of the  $A$  exciton in ML MoS<sub>2</sub> in reflectivity and PL spectroscopy are reported. These experiments allow us to estimate the exci-

ton binding energies and oscillator strengths. The importance of accounting for the light propagation in the multilayer van der Waals heterostructure for quantitative description of the experimental data is demonstrated.

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- [1] K. S. Novoselov, A. Mishchenko, A. Carvalho, and A. H. C. Neto, *Science* **353**, aac9439 (2016).
- [2] A. K. Geim and I. V. Grigorieva, *Nature (London)* **499**, 419 (2013).
- [3] K. F. Mak, C. Lee, J. Hone, J. Shan, and T. F. Heinz, *Phys. Rev. Lett.* **105**, 136805 (2010).
- [4] A. Splendiani, L. Sun, Y. Zhang, T. Li, J. Kim, C.-Y. Chim, G. Galli, and F. Wang, *Nano Lett.* **10**, 1271 (2010).
- [5] Q. H. Wang, K. Kalantar-Zadeh, A. Kis, J. N. Coleman, and M. S. Strano, *Nat. Nanotechnol.* **7**, 699 (2012).
- [6] K. F. Mak and J. Shan, *Nat. Photonics* **10**, 216 (2016).
- [7] G. Wang, A. Chernikov, M. M. Glazov, T. F. Heinz, X. Marie, T. Amand, and B. Urbaszek, *arXiv:1707.05863*.
- [8] K. He, N. Kumar, L. Zhao, Z. Wang, K. F. Mak, H. Zhao, and J. Shan, *Phys. Rev. Lett.* **113**, 026803 (2014).
- [9] M. M. Ugeda, A. J. Bradley, S.-F. Shi, F. H. da Jornada, Y. Zhang, D. Y. Qiu, S.-K. Mo, Z. Hussain, Z.-X. Shen, F. Wang *et al.*, *Nat. Mater.* **13**, 1091 (2014).
- [10] A. Chernikov, T. C. Berkelbach, H. M. Hill, A. Rigosi, Y. Li, O. B. Aslan, D. R. Reichman, M. S. Hybertsen, and T. F. Heinz, *Phys. Rev. Lett.* **113**, 076802 (2014).
- [11] Z. Ye, T. Cao, K. O'Brien, H. Zhu, X. Yin, Y. Wang, S. G. Louie, and X. Zhang, *Nature (London)* **513**, 214 (2014).
- [12] D. Y. Qiu, F. H. da Jornada, and S. G. Louie, *Phys. Rev. Lett.* **111**, 216805 (2013).
- [13] A. Ramasubramaniam, *Phys. Rev. B* **86**, 115409 (2012).
- [14] G. Wang, X. Marie, I. Gerber, T. Amand, D. Lagarde, L. Bouet, M. Vidal, A. Balocchi, and B. Urbaszek, *Phys. Rev. Lett.* **114**, 097403 (2015).
- [15] D. Xiao, G.-B. Liu, W. Feng, X. Xu, and W. Yao, *Phys. Rev. Lett.* **108**, 196802 (2012).
- [16] G. Sallen, L. Bouet, X. Marie, G. Wang, C. R. Zhu, W. P. Han, Y. Lu, P. H. Tan, T. Amand, B. L. Liu *et al.*, *Phys. Rev. B* **86**, 081301 (2012).
- [17] K. F. Mak, K. He, J. Shan, and T. F. Heinz, *Nat. Nanotechnol.* **7**, 494 (2012).
- [18] G. Kioseoglou, A. T. Hanbicki, M. Currie, A. L. Friedman, D. Gunlycke, and B. T. Jonker, *Appl. Phys. Lett.* **101**, 221907 (2012).
- [19] T. Cao, G. Wang, W. Han, H. Ye, C. Zhu, J. Shi, Q. Niu, P. Tan, E. Wang, B. Liu *et al.*, *Nat. Commun.* **3**, 887 (2012).
- [20] A. M. Jones, H. Yu, N. J. Ghimire, S. Wu, G. Aivazian, J. S. Ross, B. Zhao, J. Yan, D. G. Mandrus, D. Xiao *et al.*, *Nat. Nanotechnol.* **8**, 634 (2013).
- [21] L. Yang, N. A. Sinitsyn, W. Chen, J. Yuan, J. Zhang, J. Lou, and S. A. Crooker, *Nat. Phys.* **11**, 830 (2015).
- [22] J. R. Schaibley, H. Yu, G. Clark, P. Rivera, J. S. Ross, K. L. Seyler, W. Yao, and X. Xu, *Nat. Rev. Mater.* **1**, 16055 (2016).
- [23] R. G. Dickinson and L. Pauling, *J. Am. Chem. Soc.* **45**, 1466 (1923).
- [24] S. Dufferwiel, S. Schwarz, F. Withers, A. Trichet, F. Li, M. Sich, O. D. Pozo-Zamudio, C. Clark, A. Nalitov, D. Solnyshkov *et al.*, *Nat. Commun.* **6**, 8579 (2015).
- [25] G. Wang, X. Marie, B. L. Liu, T. Amand, C. Robert, F. Cadiz, P. Renucci, and B. Urbaszek, *Phys. Rev. Lett.* **117**, 187401 (2016).
- [26] X. Liu, T. Galfsky, Z. Sun, F. Xia, E.-C. Lin, Y.-H. Lee, S. Kéna-Cohen, and V. M. Menon, *Nat. Photonics* **9**, 30 (2015).
- [27] N. Lundt, S. Klembt, E. Cherotchenko, O. Iff, A. V. Nalitov, M. Klaas, S. Betzold, C. P. Dietrich, A. V. Kavokin, S. Höfling *et al.*, *Nat. Commun.* **7**, 13328 (2016).
- [28] L. Zhang, R. Gogna, W. Burg, E. Tutuc, and H. Deng, *arXiv:1706.08464*.
- [29] A. Srivastava, M. Sidler, A. V. Allain, D. S. Lembke, A. Kis, and A. Imamoglu, *Nat. Phys.* **11**, 141 (2015).
- [30] A. V. Stier, K. M. McCreary, B. T. Jonker, J. Kono, and S. A. Crooker, *Nat. Commun.* **7**, 10643 (2016).
- [31] G. Wang, L. Bouet, M. M. Glazov, T. Amand, E. L. Ivchenko, E. Palleau, X. Marie, and B. Urbaszek, *2D Mater.* **2**, 034002 (2015).
- [32] G. Aivazian, Z. Gong, A. M. Jones, R.-L. Chu, J. Yan, D. G. Mandrus, C. Zhang, D. Cobden, W. Yao, and X. Xu, *Nat. Phys.* **11**, 148 (2015).
- [33] K. Hao, G. Moody, F. Wu, C. K. Dass, L. Xu, C.-H. Chen, L. Sun, M.-Y. Li, L.-J. Li, A. H. MacDonald, and X. Li, *Nat. Phys.* **12**, 677 (2016).
- [34] B. Zhu, X. Chen, and X. Cui, *Sci. Rep.* **5**, 9218 (2015).

- [35] A. Chernikov, A. M. van der Zande, H. M. Hill, A. F. Rigosi, A. Velauthapillai, J. Hone, and T. F. Heinz, *Phys. Rev. Lett.* **115**, 126802 (2015).
- [36] H. M. Hill, A. F. Rigosi, C. Roquelet, A. Chernikov, T. C. Berkelbach, D. R. Reichman, M. S. Hybertsen, L. E. Brus, and T. F. Heinz, *Nano Lett.* **15**, 2992 (2015).
- [37] B. R. Carvalho, L. M. Malard, J. M. Alves, C. Fantini, and M. A. Pimenta, *Phys. Rev. Lett.* **114**, 136403 (2015).
- [38] Y. Li, A. Chernikov, X. Zhang, A. Rigosi, H. M. Hill, A. M. van der Zande, D. A. Chenet, E.-M. Shih, J. Hone, and T. F. Heinz, *Phys. Rev. B* **90**, 205422 (2014).
- [39] K. L. Seyler, J. R. Schaibley, P. Gong, P. Rivera, A. M. Jones, S. Wu, J. Yan, D. G. Mandrus, W. Yao, and X. Xu, *Nat. Nanotechnol.* **10**, 407 (2015).
- [40] P. Soubelet, A. E. Bruchhausen, A. Fainstein, K. Nogajewski, and C. Faugeras, *Phys. Rev. B* **93**, 155407 (2016).
- [41] B. Chakraborty, H. Matte, A. Sood, and C. Rao, *J. Raman Spectrosc.* **44**, 92 (2013).
- [42] M. M. Glazov, L. E. Golub, G. Wang, X. Marie, T. Amand, and B. Urbaszek, *Phys. Rev. B* **95**, 035311 (2017).
- [43] A. Kormányos, V. Zólyomi, N. D. Drummond, P. Rakyta, G. Burkard, and V. I. Fal'ko, *Phys. Rev. B* **88**, 045416 (2013).
- [44] J. A. Miwa, S. Ulstrup, S. G. Sørensen, M. Dendzik, A. G. Čabo, M. Bianchi, J. V. Lauritsen, and P. Hofmann, *Phys. Rev. Lett.* **114**, 046802 (2015).
- [45] A. Molina-Sánchez, M. Palumbo, A. Marini, and L. Wirtz, *Phys. Rev. B* **93**, 155435 (2016).
- [46] T. Korn, S. Heydrich, M. Hirmer, J. Schmutzler, and C. Schüller, *Appl. Phys. Lett.* **99**, 102109 (2011).
- [47] D. Lagarde, L. Bouet, X. Marie, C. R. Zhu, B. L. Liu, T. Amand, P. H. Tan, and B. Urbaszek, *Phys. Rev. Lett.* **112**, 047401 (2014).
- [48] H. Zeng, J. Dai, W. Yao, D. Xiao, and X. Cui, *Nat. Nanotechnol.* **7**, 490 (2012).
- [49] A. V. Stier, N. P. Wilson, G. Clark, X. Xu, and S. A. Crooker, *Nano Lett.* **16**, 7054 (2016).
- [50] A. A. Mitioglu, K. Galkowski, A. Surrente, L. Kłopotowski, D. Dumcenco, A. Kis, D. K. Maude, and P. Plochocka, *Phys. Rev. B* **93**, 165412 (2016).
- [51] F. Cadiz, E. Courtade, C. Robert, G. Wang, Y. Shen, H. Cai, T. Taniguchi, K. Watanabe, H. Carrere, D. Lagarde *et al.*, *Phys. Rev. X* **7**, 021026 (2017).
- [52] T. Taniguchi and K. Watanabe, *J. Cryst. Growth* **303**, 525 (2007).
- [53] Z. Wang, L. Zhao, K. F. Mak, and J. Shan, *Nano Lett.* **17**, 740 (2017).
- [54] O. A. Ajayi, J. V. Ardelean, G. D. Shepard, J. Wang, A. Antony, T. Taniguchi, K. Watanabe, T. F. Heinz, S. Strauf, X.-Y. Zhu *et al.*, *2D Mater.* **4**, 031011 (2017).
- [55] M. Manca, M. M. Glazov, C. Robert, F. Cadiz, T. Taniguchi, K. Watanabe, E. Courtade, T. Amand, P. Renucci, X. Marie *et al.*, *Nat. Commun.* **8**, 14927 (2017).
- [56] C. Jin, J. Kim, J. Suh, Z. Shi, B. Chen, X. Fan, M. Kam, K. Watanabe, T. Taniguchi, S. Tongay *et al.*, *Nat. Phys.* **13**, 127 (2017).
- [57] H. Gibbs, G. Khitrova, and S. Koch, *Nat. Photonics* **5**, 273 (2011).
- [58] F. Kreller, M. Lowisch, J. Puls, and F. Henneberger, *Phys. Rev. Lett.* **75**, 2420 (1995).
- [59] J. Kasprzak, M. Richard, S. Kundermann, A. Baas, P. Jeambrun, J. Keeling, F. Marchetti, M. Szymańska, R. Andre, J. Staehli *et al.*, *Nature (London)* **443**, 409 (2006).
- [60] P. Gong, H. Yu, Y. Wang, and W. Yao, *Phys. Rev. B* **95**, 125420 (2017).
- [61] Y. K. Luo, J. Xu, T. Zhu, G. Wu, E. J. McCormick, W. Zhan, M. R. Neupane, and R. K. Kawakami, *Nano Lett.* **17**, 3877 (2017).
- [62] Z. Y. Zhu, Y. C. Cheng, and U. Schwingenschlögl, *Phys. Rev. B* **84**, 153402 (2011).
- [63] T. Cheiwchanchamnangij and W. R. L. Lambrecht, *Phys. Rev. B* **85**, 205302 (2012).
- [64] See Supplemental Material at <http://link.aps.org/supplemental/10.1103/PhysRevMaterials.2.011001> for additional information on experimental methods, optical spectroscopy results, and modeling.
- [65] G. Wang, M. M. Glazov, C. Robert, T. Amand, X. Marie, and B. Urbaszek, *Phys. Rev. Lett.* **115**, 117401 (2015).
- [66] M. M. Glazov, T. Amand, X. Marie, D. Lagarde, L. Bouet, and B. Urbaszek, *Phys. Rev. B* **89**, 201302 (2014).
- [67] A. Molina-Sánchez and L. Wirtz, *Phys. Rev. B* **84**, 155413 (2011).
- [68] C. M. Chow, H. Yu, A. M. Jones, J. R. Schaibley, M. Koehler, D. G. Mandrus, R. Merlin, W. Yao, and X. Xu, *npj 2D Mater. Appl.* **1**, 33 (2017).
- [69] K. F. Mak, K. He, C. Lee, G. H. Lee, J. Hone, T. F. Heinz, and J. Shan, *Nat. Mater.* **12**, 207 (2013).
- [70] J. S. Ross, S. Wu, H. Yu, N. J. Ghimire, A. M. Jones, G. Aivazian, J. Yan, D. G. Mandrus, D. Xiao, W. Yao *et al.*, *Nat. Commun.* **4**, 1474 (2013).
- [71] N. S. Rytova, *Proc. MSU, Phys., Astron.* **3**, 30 (1967).
- [72] L. V. Keldysh, *Soviet Journal of Experimental and Theoretical Physics Letters* **29**, 658 (1979).
- [73] P. Cudazzo, I. V. Tokatly, and A. Rubio, *Phys. Rev. B* **84**, 085406 (2011).
- [74] T. C. Berkelbach, M. S. Hybertsen, and D. R. Reichman, *Phys. Rev. B* **88**, 045318 (2013).
- [75] F. Wu, F. Qu, and A. H. MacDonald, *Phys. Rev. B* **91**, 075310 (2015).
- [76] A. V. Stier, N. P. Wilson, K. A. Velizhanin, J. Kono, X. Xu, and S. A. Crooker, [arXiv:1709.00123](https://arxiv.org/abs/1709.00123).
- [77] M. Shinada and S. Sugano, *J. Phys. Soc. Jpn.* **21**, 1936 (1966).
- [78] D. R. Kazanov, A. V. Poshakinskiy, S. V. Sorokin, and T. V. Shubina, *J. Phys.: Conf. Ser.* **917**, 062022 (2017).
- [79] D.-H. Lien, J. S. Kang, M. Amani, K. Chen, M. Tosun, H.-P. Wang, T. Roy, M. S. Eggleston, M. C. Wu, M. Dubey *et al.*, *Nano Lett.* **15**, 1356 (2015).
- [80] E. L. Ivchenko, *Optical Spectroscopy of Semiconductor Nanostructures* (Alpha Science, Harrow, UK, 2005).